Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 NANOCRYSTALLIZATION EFFECTS ON THE SPECIFIC HEAT OF Fe-Co-Nb-B

AMORPHOUS ALLOY

J.S. Blázquez^a, M. Millán^a, C.F. Conde^a, V. Franco^a, A. Conde^a^{*},

S. Lozano-Pérez^b

^a Dpto. Física de la Materia Condensada, ICMSE-CSIC, Universidad de Sevilla, P.O. Box 1065, 41080, Sevilla, Spain

^b Department of Materials, University of Oxford, Parks Road, Oxford, OX1 3PH, UK.

Abstract

Specific heat at constant pressure, C_P , was measured on amorphous, nanocrystalline and fully crystalline samples of Fe₆₀Co₁₈Nb₆B₁₆ alloy. Magnetic and calorimetric measurements agree, describing a continuously decreasing Curie temperature of the amorphous phase. A clear enhancement of C_P over the Dulong-Pettit limit has been observed (from 14 to 25 %). Part of the enhancement is due to magnetic (mainly amorphous phase) and electronic contributions, although an excess volume can be inferred from the high value of the slope of C_P versus temperature.

PACS codes: 61.46.Hk; 65.40.Ba.

Keywords: Specific heat; nanocrystalline alloys

Nanocrystalline materials are characterized by an important fraction of atoms located at the

^{*} Prof. A. Conde:Tel: (34) 95 455 28 85, Fax: (34) 95 461 20 97, e-mail: conde@us.es

Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 interfaces, which can yield important differences in their properties with respect to those of conventional systems. These atoms can be considered as if they were in a dilated lattice leading to the definition of an excess volume [1], V_X , which affects several mechanical and thermal properties, e.g. enhancement of specific heat at constant pressure, C_P . This enhancement has been observed for nanocrystalline, quasicrystalline and amorphous structures [1,2,3]. In this paper, measurements of C_P for amorphous, nanocrystalline and fully crystalline Fe₆₀Co₁₈Nb₆B₁₆ alloy are reported.

Melt-spun amorphous ribbons were devitrified up to different crystalline volume fraction, X_C, by heating at 20 K/min in a Perkin-Elmer DSC7, differential scanning calorimeter (DSC). The devitrification of this alloy (see inset of Fig. 1) occurs through two exothermic processes, the former ascribed to the nanocrystallization of α -FeCo phase and the latter to the completion of devitrification (mainly α -Fe and Fe₂₃B₆ type phases). X_C was calculated from the area below the DSC peak between the onset and the maximum temperature achieved during the corresponding heating treatment, taking into account that the end of the nanocrystallization corresponds to X_C=0.54 [4]. DSC was also used to measure the specific heat at constant pressure, C_P, following the procedure described elsewhere [5]. The selected temperature range was low enough to prevent any thermal evolution of the samples. High resolution electron microscopy (HREM) was performed in a Jeol JEM3000F operated at 297 kV. Specific saturation magnetization, σ_S , was measured in a vibrating sample magnetomer (VSM), extrapolating to zero field the magnetization curves obtained under a maximum

2

Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 applied field of 1.5 T.

Two different temperature ranges were chosen for C_P measurements: a first one from 523 to 723 K and a second one from 523 to 573 K. Figure 1(above) shows the C_P values obtained in a continuous scan for three studied samples: relaxed amorphous, nanocrystalline and fully crystalline samples. For amorphous samples, C_P curves show a clear and broad maximum at 662 K ascribed to the Curie transition of the amorphous phase. The broadening of this peak is related to experimental factors [6] (e.g. temperature gradients in the samples). As nanocrystallization progresses this maximum becomes weaker, being undetected for crystalline volume fractions above ~0.45. Excellent agreement is found with T_C obtained from the derivative of the $\sigma_S(T)$ plots (figure 2).

Reducing the temperature range in the second set of measurements, errors due to baseline drift are minimized. Figure 1(below) shows the C_P at 550 K as a function of X_C. The monotonous decrease was confirmed from a different set of measurements in which the reference furnace of the DSC is filled with the sample heated at an immediately lower temperature instead of an empty pan. A clear enhancement of 25 % in the C_P value with respect to the Dulong-Pettit limit (DP~485 JK⁻¹kg⁻¹) is observed. This enhancement, Δ C_P decreases as the nanocrystallization progresses saturating at ~14 % for nanocrystalline samples with X_C>0.45.

The value of specific heat at constant volume in the harmonic approximation, C_{VH}, at 550 K is close to DP limit, using the Debye temperature values found for similar alloys (~350 K)

Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075

[7]. For conventional materials, the constant volume specific heat, C_V , results from addition of different contributions [6]: anharmonic vibrations (C_{VA}), conducting electrons (C_{VE}), equilibrium vacancies in the lattice (C_{VV}), and magnetic contribution (C_{VM}). Moreover, $C_P=C_V+C_D$, where C_D is the dilatation correction. In the following discussion, C_{VV} and C_{VA} will be neglected (temperature is far from the melting point [8]). C_D and C_{VE} linearly increase with temperature. The experimental slope of the C_P curves far from the Curie transition ($m_{CP}=m_{CVE}+m_{CD}\sim 0.25 \text{ J kg}^{-1}\text{K}^{-2}=0.013 \text{ J mol}^{-1}\text{K}^{-2}$) is almost independent of X_C . Considering $m_{CVE}\sim 0.005 \text{ J mol}^{-1}\text{K}^{-2}$ (pure Fe [9]), the expression $C_D = (C_{FH} \cdot \gamma)^2 / B \cdot V$ [1,6] yields a bulk modulus B~50 GPa, where $\gamma \sim 2$ [10] is the Grüneissen parameter and V the molar volume. This value of B would imply an excess volume ~0.2 [1], ascribed to the nanocrystalline structure. HREM (Fig. 3) was used to locally measure interatomic distances. Results indicate an increase in the lattice parameter near the surface of the nanocrystals compared to the core region (<8%), although the derived error for ΔV is too large, preventing a quantitative estimation.

 C_{VM} contributes to the ΔC_P mainly due to the amorphous phase magnetic transition and C_{VM} of α -FeCo can be neglected as its T_{C} ~1200 K. Therefore, ΔC_P may decrease as nanocrystallization progresses, as observed. Finally, fully crystalline sample exhibits a similar C_P value than nanocrystalline ones. This could be due to the Curie transition of Fe₂₃B₆-type phase, slightly above the studied temperature range [11].

Specific heat enhancement over Dulong-Pettit limit has been observed on Fe₆₀Co₁₈Nb₆B₁₆

4

Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 amorphous and nanocrystalline alloys. Part of the enhancement is due to magnetic (mainly amorphous phase) and electronic contributions. An excess volume of ~0.2 could be estimated. Magnetic and calorimetric measurements agree, describing a continuously decreasing Curie temperature of the amorphous phase.

This work was supported by MEC of the Spanish Government and EU FEDER (Project MAT2007-65227), the PAI of the Regional Government of Andalucía (Project P06-FQM-01823). The TEM investigations were supported by the IP3 project of the 6th Framework

Programme of the European Commission: ESTEEM Contract number 026019. J.S.B.

acknowledges a research contract from this Regional Government.

Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075

Figure captions

Figure 1. Above: C_P curves at 20 K/min for samples pre-heated at the indicated temperatures.

Inset: DSC scan at 20 K/min of as-cast alloy indicating the heating temperatures used.

Below: C_P values at 550 K. The Dulong- Pettit limit is 485 J kg⁻¹K⁻¹.

Figure 2. σ_S curves for the different studies samples. Inset: T_C of the amorphous phase as a

function of X_{C} from VSM and C_{P} measurements.

Figure 3. HREM image of a nanocrystal at [100] zone axis for the sample heated up to 813 K.

Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 Figure 1



Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 Figure 2



Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 Figure 3



Journal of Non Crystalline Solids. Vol. 354. Núm. 47-51. 2008. Pag. 5135-5137 http://dx.doi.org/10.1016/j.jnoncrysol.2008.05.075 References

- [1] M. Wagner, Phys. Rev. B 45 (1992) 635-639.
- [2] G. Kh. Panova, N. A. Chernoplekov, A. A. Shikov, T. Kemeny, L. F. Kiss, Phys. Solid State 49 (2007) 1617-1622.
 [3] L. Wang, Z. Tan, S. Meng, A. Druzhinina, R. A. Varushchenko, G. Li, J. Non-Cryst. Sol. 296 (2001) 139-142.
- [4] J. S. Blázquez, V. Franco, C. F. Conde, A. Conde, J. Mag. Mag. Mat. 254-255 (2003) 460-462.
- [5] Perkin-Elmer DSC2 Operation Instructions (1981) Norwalk, Connecticut, U.S.A. p. 3-20.
 [6] P. J. Meschter, J. W. Wright, C. R. Brooks, T. G. Kollie, J. Phys. Chem. Solids 42 (1981) 861-871.
- [7] J. Z. Jiang, J. Mag. Mag. Mat. 154 (1996) 375-381
 [8] P. Mazur, J. Appl. Phys. 40 (1969) 482-484.
- [9] N. W. Ashcroft, I. Mermin, N. David, Solid State Physics, Ed. W. D. Saunders (1976) U.S.A. p. 48.
- [10] C. Dai, H. Tan, H. Geng, J. Appl. Phys. 92 (2002) 5019-5026.
 [11] J. S. Blázquez, C. F. Conde, A. Conde, J. Non-Cryst. Sol. 287 (2001) 187-192.