

1 Heteroatom Framework Distribution and Layer Charge of 2 Sodium Taeniolite

3 Ana C. Perdigón¹, Carmen Pesquera¹, Agustín Cota², Francisco J. Osuna³, Esperanza
4 Pavón³, María D. Alba^{3*}

5 ¹ Departamento de Química e Ingeniería de Procesos y Recursos. Universidad
6 Cantabria. Avda. Los Castros s/n. 39005 Santander (Spain)

7 ² Laboratorio de Rayos-X. Centro de Investigación, Ciencia y Tecnología de la
8 Universidad de Sevilla (CITIUS). Avda. Reina Mercedes, 4. 41012 Sevilla (Spain)

9 ³ Instituto Ciencia de los Materiales de Sevilla (CSIC-US). Avda. Reina Mercedes, 49.
10 41092 Sevilla (Spain)

11

12 ABSTRACT

13 The most advanced applications of clays depend crucially on their hydration state and
14 swelling is probably the most important feature of expandable 2:1 layered **silicate**.
15 Sodium Taeniolite, Na-TAE, a swelling trioctahedral fluormica, has been synthesized
16 and studied using thermogravimetric analysis, X-ray diffraction, scanning electron
17 microscopy and infrared and solid state NMR spectroscopies. **The results indicated the**
18 **formation of a swelling 2:1 phyllosilicate with actual layer charge lower than the**
19 **nominal one. Herein, a new heteroatom distribution and more accurate composition**
20 **could be deduced.**

21

22

23

24

25

26

27 **1. Introduction**

28 Swelling silicates of the 2:1 phyllosilicate family with low negative charge are
29 2D host materials with multiple and diverse physicochemical applications going from
30 the well-known catalysis (Alves et al., 2014; Breu et al., 1999; Centi and Perathoner,
31 2008; Cheng, 1999; Thomas, 1988; Vaccari, 1999) and adsorption/separation (Barrer,
32 1989; Mercier and Pinnavaia, 1998; Okada and Ogawa, 2017; Tomohiko et al., 2014) to
33 the latest nanotechnology (Schollhorn, 1996; Schoonheydt and Umemura, 2017), sensor
34 technology (Mallouk and Gavin, 1998) and optoelectronics.(Kunz et al., 2013; Lezhnina
35 et al., 2007).

36 Chemical structure is the ultimate responsible of those interesting applications.
37 The structure, an octahedral sheet sandwiched between two tetrahedral sheets, is
38 extremely flexible to isomorphic substitutions. Additionally, the interlayer cation can be
39 substituted by cation exchange reactions and large inorganic or organic molecules can
40 be introduced in the interlayer space giving those materials a wide adaptability.

41 Chemical reactivity and stability in the interlayer space as a reaction media
42 strongly depends not only of the layer charge but also from its origin (Alba et al., 2001a,
43 b; Alba et al., 2010; Alba et al., 2009; Chain et al., 2013). On the one hand, cation
44 substitutions on tetrahedral sheet favor reconstructive reactions under subcritical
45 conditions (Alba et al., 2001a). On the other hand, cation substitutions on octahedral
46 sheet provides an excess of negative charge that is delocalized in the framework
47 (Sposito and Prost, 1982), and, thus, a more homogeneous distribution of cations in the
48 interlayer space. Moreover, the hydration properties of 2:1 phyllosilicates are also
49 governed by those structural features because their swelling behaviour depends on the
50 repulsive forces relating to the interactions between the 2:1 layers and the attractive

51 forces between the interlayer cation and the negative charged siloxane surface (Pavon et
52 al., 2014).

53 True and brittle natural micas constitute a major group of those 2D host
54 materials but layer charge is high and strong Coulomb interactions with interlayer
55 cations usually inhibit their swelling and hence, a very low cation-exchange capacity
56 (CEC) results and limits applicability. However, as will be shown in the following,
57 swelling high-charge micas can be easily synthesized and their CEC tuned via the
58 degree of isomorphous substitution, broadening their wide-ranging uses.

59 Depending on the origin of the layer charge, two family of swelling mica can be
60 found. First, synthetic fluorine micas have been recently synthesized (Alba et al., 2006)
61 that are able to swell despite their high layer charge originated in the tetrahedral sheet
62 by Si/Al replacements (Alba et al., 2006; Naranjo et al., 2015; Pavon et al., 2013).
63 Consequently, those materials, referred as Na-*n*-Mica or Na-Mica-*n* (being *n* the layer
64 charge and related to the ratio Si/Al) have potential used as decontaminants and storage
65 media (Alba et al., 2006; Garcia-Jimenez et al., 2016; Gregorkiewitz and Rausell-
66 Colom, 1987; Park et al., 2002; Paulus et al., 1992).

67 Second, synthetic sodium Taeniolite, an analogous of the swelling fluoromica
68 Na-Mica-2, where substitutions in the octahedral sheet (Mg/Li) are the ultimate
69 responsible of layer charge (Kitajima et al., 1985; Kitajima and Takusagawa, 1990;
70 Kitajima et al., 1991). Sodium Taeniolite ($\text{NaSi}_4\text{Mg}_2\text{LiO}_{10}\text{F}_2 \cdot x\text{H}_2\text{O}$) exhibits reversible
71 swelling (Toraya et al., 1977) and Kitajima *et al.* (Kitajima et al., 1973) explained its
72 swelling characteristics by the high hydration energy of the interlayer sodium ions.
73 Taeniolite derivatives are interesting microporous host materials that can be used as
74 sensors for redox-active ambient gas (Baumgartner et al., 2008; Mariychuk et al., 2007).

75 The capability of the 2:1 structure to vary the framework chemical composition
76 and heteroatom distribution is crucial to design new materials with desired properties;
77 however, it leaves one worrying about the homogeneity of the synthesized material. In
78 many cases, the charge densities vary not only from one silicate layer to other in a
79 crystallite (Lagaly, 1994) but also from domain to domain within a single silicate layer
80 (Muller et al., 1997) and this heterogeneous distribution of negative charge influences
81 the distribution cations in the interlayer space (Breu et al., 1999).

82 The charge density of the silicate layers is of utmost importance for the
83 properties of the material. Na-Mica-n have been extensively analysed (Alba et al., 2006)
84 but, to our knowledge, the only available information of Na-Taeniolite is related to their
85 crystalline order and swelling capacity (Kitajima et al., 1985; Kitajima et al., 1973;
86 Miyake et al., 1993; Moore et al., 1997; Toraya et al., 1977). Thus, the goal of this
87 research was to shed some light on the uniformity of their composition, charge density,
88 and interlayer composition by the combination of characterization techniques that
89 provide structural information at long and short range.

90

91 **2. Experimental section**

92 *2.1. Synthesis method.*

93 The reagents of SiO₂ (Sigma, 99.8% CAS N° 112945-52-5), LiF (Aldrich, 99.99%, CAS
94 N° 1309-48-4), MgO (Aldrich, 99.9%, CAS N° 1309-48-4) and NaF (Aldrich, 99.0%,
95 CAS N° 7681-49-4) were dried at 100 °C for 24 h and carefully weighed out according
96 to the formulae [Na₂][Si₈][Mg₂₄Li₂]O₂₀F₄.

97 All reagents were mixed and vigorously ground before heating at 1100 °C in a Pt
98 crucible for 2 h. After cooling up to 950 °C, the solid was quenched to room

99 temperature (RT) and equilibrated at RT ambient humidity. The as-synthesized sample
100 was named as Na-TAE.

101

102 *2.2.Characterization.*

103 TG/DTA experiment was carried out using a TA (model STD-Q600) instrument, in
104 Characterization Service (CITIUS, University of Seville, Spain), with alumina as
105 reference. The sample was placed into a Pt crucible and maintained at air throughout the
106 heating period. The temperature was increased at a constant rate of 10° C/min.

107 Powder X-ray diffraction (XRD) was performed at the X-ray laboratory (CITIUS,
108 University of Seville, Spain) on a Bruker D8 Advance Bragg-Brentano instrument
109 equipped with a Cu K α radiation source operating at 40 kV and 40 mA. The
110 diffractogram was obtained in the 2 θ range of 3–70° with a step size of 0.015° and a
111 step time of 0.1 s.

112 FTIR spectrum was recorded in the range 4000–300 cm⁻¹ by the Spectroscopy
113 Service of the ICMS (CSIC-US, Seville, Spain), as KBr pellets, using a Nicolet
114 spectrometer (model 510P) with a nominal resolution of 4 cm⁻¹.

115 The morphology of the sample was analyzed by Scanning Electron Microscopy,
116 SEM, (JEOL, Model JSM 5400) at 20kV in the Microscopy Service of ICMS (CSIC-
117 US). An EDX system (Oxford Link ISIS) was fitted to the SEM equipment to perform
118 chemical analyses of the sample using a Si/Li detector with Be window.

119 Single-pulse (SP) MAS NMR spectra were recorded in the Nuclear Magnetic
120 Resonance Unit at the University of Cordoba (Spain) on a Bruker AVANCE WB400
121 spectrometer equipped with a multinuclear probe. Powdered sample was packed in 3.2
122 mm zirconia rotors and spun at 10 kHz. ²⁹Si MAS NMR spectrum was acquired at a
123 frequency of 79.49 MHz, using a pulse width of 2.7 μ s ($\pi/2$ pulse length = 7.1 μ s) and

124 delay times of 3 s. ^6Li MAS NMR spectrum was recorded at 58.86 MHz with a pulse
125 width of 0.9 μs ($\pi/2$ pulse length = 1.8 μs) and a delay time of 30 s, in this case the rotor
126 was spun at 14 kHz to improve the spectral resolution. ^{23}Na MAS NMR spectrum was
127 recorded at 105.84 MHz with a pulse width of 0.75 μs ($\pi/2$ pulse length = 4.5 μs) and a
128 delay time of 0.1 s. The ^{19}F MAS spectrum was obtained using typical $\pi/2$ pulse widths
129 of 2.9 μs and a pulse space of 2 s. The chemical shift values were reported in ppm from
130 tetramethylsilane for ^{29}Si , from NaF for ^{19}F and from a 0.1 M LiCl and NaCl solution
131 for ^6Li and ^{23}Na , respectively. Spectra were simulated using the DMFIT software
132 (Massiot et al., 2002). A Gaussian-Lorentzian model was used for all peaks, and
133 linewidth, position and amplitude were the fitted parameters.

134

135 **3. Results and discussion**

136 The XRD pattern (Fig. 1) is similar to the previously reported for the one-layer
137 hydrated 2:1 structure of Na-Taeniolite (Miyake et al., 1993; Suzuki et al., 2008;
138 Yamaguchi et al., 2012) and showed a sharp $00l$ diffraction peak at $7.2^\circ 2\theta$ ($d_{001}=12.2$
139 \AA) and high order $00l$ reflections corresponding to the single-layer hydrate state of the
140 interlayer cations (Kitajima et al., 1985). Kalo et al. (Kalo et al., 2010) also observed
141 that $\text{Na}_{0.6}$ -fluorohectorite spontaneously absorbs air moisture to form a monolayer
142 hydrate (ca. 2 H_2O per unit cell) at ambient conditions. Additionally, some reflections
143 due to protoamphibole (PDF 00-013-0409), an orthorhombic fluoroamphibole, (Gibbs et
144 al., 1960) are detected in the XRD patterns, as previously reported for Na-
145 fluorohectorite (Kitajima et al., 1985). Also, Yamaguchi et al. (Yamaguchi et al., 2012)
146 observed some impurities in Taeniolite obtained from Na-rich raw batches (≥ 3 mol
147 NaCl) that they could not identify and nor remove after rigorous washing.

148 Fig. S1 showed the c-axis projection of the K-Taeniolite structure (Toraya et al.,
149 1977) and the a-axis projection of protoamphibole structure (Gibbs, 1969), the
150 similarity of building blocks of both, protoamphibole being slightly denser, may favour
151 that protoamphibole acts as a natural competitor for the 2:1 layer structure (Breu et al.,
152 2001).

153 The hydration state of the Na-TAE was tested by thermogravimetric analysis
154 (Fig. S2). In the region between 25 °C and 150 °C, where water molecules are desorbed,
155 a unique weight loss was observed with the maximum in the DTG curve at 66 °C,
156 associated to a 4.92 % of weight loss. If the ideal formula, $\text{Na}_2\text{Si}_8\text{Mg}_4\text{Li}_2\text{O}_{20}\text{F}_4 \cdot x\text{H}_2\text{O}$, is
157 taken into account this weight loss corresponds to 2.23 H_2O per unit cell. This value is
158 low as previously reported for Na-Taeniolite (Miyake et al., 1993), $\text{Na}_{0.6}$ -fluorohectorite
159 (Kalo et al., 2010) and for Na-Mica-2, a fluoromica with layer charge originated in the
160 tetrahedral sheet instead of in the octahedral one (Gregorkiewitz and Rausell-Colom,
161 1987; Kalo et al., 2013; Pavon et al., 2013). **The low water weight loss is justified by**
162 **the coordination of the interlayer cation to the basal oxygen of the tetrahedral sheet and,**
163 **maybe, fluorine** (Rayner-Canham and Overton, 2006).

164 The morphology of the sample was analysed by scanning electron microscopy
165 (Fig. 2) which shows that most particles have a lamellar structure compatible with the
166 2:1 phyllosilicate, Na-TAE, (image a). **The EDX spectrum (spectrum a) shows lines**
167 **compatible with chemical composition of Na-TAE but the semiquantitative analysis of**
168 **the peaks intensities reveals that Mg/Si and Na/Si ratios are lower than that expected for**
169 **Na-TAE-2 (Mg/Si=0.35 vs 0.50 and Na/Si=0.07 vs 0.25). Particles with a denser**
170 morphology (image b) was observed that can be attributed to the protoamphibole
171 detected by XRD and its EDX spectrum shows lines compatible with its chemical
172 composition (spectrum b).

173 Once confirmed the hydrated laminar structure of Na-TAE, the tetrahedral lattice
174 vibrations were analysed by IRFT spectroscopy (Fig. 3) because they are affected by the
175 magnitude of the layer charge and/or by the changes in octahedral and interlayer
176 compositions (Kitajima et al., 1985; Kitajima and Takusagawa, 1990; Kitajima et al.,
177 1991). The IRFT spectrum in this region showed absorption bands at ca. 980 and 1102
178 cm^{-1} assigned to e_1^1 and a_1^1 , stretching Si-O_a, mode (Kitajima et al., 1985; Kitajima and
179 Takusagawa, 1990) and a band at ca. 715 cm^{-1} due to a_1^2 mode (Yamaguchi et al.,
180 2012). Consequently, judging from the IR spectra, the Na-TAE has similar chemical
181 composition reported in the literature for Na-Taeniolite (Yamaguchi et al., 2012).

182 The local structure of the heteroatoms in the framework and in the interlayer
183 space was studied by MAS NMR of the ^{29}Si , ^{19}F , ^6Li and ^{23}Na nuclei (Fig. 4); the
184 spectra were deconvoluted and the fitting parameters (Table 1) were assigned to
185 structural features.

186 The ^{29}Si MAS NMR spectrum shows a main signal at -91.37 ppm and two minor
187 signals at -85.07 and -83.39 ppm. The main signal resonates in the region of Q³(0Al)
188 (Sanz et al., 2006) as due to the 2:1 phyllosilicate (Na-TAE) and the minors
189 contributions resonated in the region of Q² and is compatible with Si in protoamphibole
190 (Welch et al., 1998), the impurity observed by XRD. The percentage of the area of the
191 peaks (Table 1) indicate **that only a 5% of the Si atoms corresponds to protoamphibole.**

192 The ^{19}F spectrum is characterized by three signals at -177.39, -183.53 and -
193 189.44 ppm, corresponding to F(Mg-Mg-Mg), F(Mg-Mg-Li) and F(Mg-Li-Li),
194 respectively (Huve et al., 1992; Kaviratna and Pinnavaia, 1994; Keenan et al., 2013;
195 Labouriau et al., 1995). The spectral deconvolution is performed to obtain quantitative
196 information of the relative $[\text{Mg}^{2+}]_3$ to $[\text{Mg}^{2+}][\text{Li}^+]_2$ environmental populations (see
197 Table 1). From signal intensities, a $\text{Mg}^{2+}/\text{Li}^+$ ratio equals to 3.20 is calculated and it is

198 representative of a layer charge equals to ca. -1.43. Therefore, the formulae would be
199 approximately rewritten as $M_{1.43}[Si_8][Mg_{4.57}Li_{1.43}]O_{20}F_4$, where M=Na and/or Li.

200 The assumption of a layer charge lower than the nominal layer charged can be
201 corroborated by a deep analysis of the signals' chemical shifts from the ^{29}Si MAS NMR
202 spectrum (Komarneni et al., 2005). As previously reported in the literature, ^{29}Si
203 chemical shift in 2:1 phyllosilicates is quite sensible to layer charge, origin of the charge
204 and the anion of octahedral coordination sphere (OH or F) (Alba et al., 2006; Keenan et
205 al., 2013; Sanz et al., 2006). The correlation between chemical shift (δ) and the absolute
206 value of the layer charge (x) in Fig. 5 uses data from literature (Alba et al., 2006;
207 Gerstmans et al., 2008; Keenan et al., 2013; Sanz et al., 2006) and analyses structural
208 features. The anion nature of the octahedral sheet, OH or F, (filled star vs filled circle);
209 the nature of the octahedral cation (pink vs purple stars and blue vs cyan circles) and the
210 nature of the interlayer cation (blue circle vs. blue triangle and purple star vs. purple
211 triangle) are analyzed. The relation between the chemical shift and absolute value of
212 layer charge depends mainly on the first two factors, therefore, four grouped data can be
213 set up and four linear correlations calculated:

214 Line I: $\delta = -98.25 + 2.88 \cdot x$, $R^2 = 0.983$ (OH and Mg in octahedral sheet)

215 Line II: $\delta = -98.29 + 2.66 \cdot x$, $R^2 = 0.911$ (OH and Mg-Al in octahedral sheet)

216 Line III: $\delta = -97.52 + 3.61 \cdot x$, $R^2 = 0.942$ (F and Mg-Li in octahedral sheet)

217 The set of chemical shift fitting with Line III corresponds to 2:1 phyllosilicate
218 with the most similar composition to Na-TAE. Consequently, this equation has been
219 employed for a semiquantitative estimation of the layer charge of Na-TAE, returning a
220 value ≤ -1.70 , lower than the nominal value of -2. In the literature (Kitajima and
221 Daimon, 1972; Kondo et al., 1980; Yamaguchi et al., 2012), a layer charge equal to -2
222 was assumed on the base of the nominal stoichiometry of the reaction mixture. In this

223 sense, Kitajima et al. (Kitajima et al., 1991) observed that the stretching Si-O_a in FTIR
224 spectra shifted to lower frequency as layer charge increased, however, Na-Taeniolite
225 reported in literature showed a slightly higher frequency (1110 cm⁻¹) (Yamaguchi et al.,
226 2012) than Na-TAE reported here (1102 cm⁻¹), then those Na-Taeniolites could also
227 exhibit a layer charger lower than -2.

228 Once known the actual framework heteroatom distribution, the analysis of the
229 interlayer space has been performed thorough ²³Na and ⁶Li MAS NMR. ²³Na MAS
230 NMR spectroscopy is sensitive to the local environment and will not only be influenced
231 by the hydration, but also by different planar defects in the stacking of the 2:1 layers
232 (Moller et al., 2010). The ²³Na spectrum is characterized by a broad band that has been
233 deconvoluted in four peaks (Table 1) and they correspond to one-layer hydrate (78.4 %)
234 and **poor hydrated** exchangeable sodium (21.6%) (Naranjo et al., 2015). It should be
235 noted that ²³Na is a quadrupolar nuclei with a half-integer spin and, therefore, affected
236 by second order quadrupolar interaction. Nevertheless, it was almost eliminated by the
237 rotor spun and the magnetic field and very small spinning side band are observed.

238 Finally, ⁶Li MAS NMR spectrum is the overlapping of two resonances at 0.23
239 and -0.72 ppm, **which can be due to hydrated interlayer lithium cation and lithium in the**
240 **octahedral sheet of trioctahedral 2:1 phyllosilicates. Bond et al. (Bond et al., 1991)**
241 **observed a unique signal in Na-Laponite at -0.73 ppm due to Li in octahedral sheet and**
242 **two signals when Na was exchanged by Li, being the new resonance more deshielded.**
243 It is remarkable that no resonance associated to poor hydrated lithium is observed and
244 could be explained because of the tendency to form inner sphere complex for sodium
245 (lower hydration state) is higher than that of lithium (Pavon et al., 2013). The proportion
246 found between both location of lithium (Table 1), **the ¹⁹F MAS NMR and TG results**

247 allow to estimate a Na-TAE composition such as:
248 $\text{Li}_{0.91}\text{Na}_{0.52}[\text{Si}_8][\text{Mg}_{4.57}\text{Li}_{1.43}]\text{O}_{20}\text{F}_4 \cdot 2.18\text{H}_2\text{O}$.

249

250 **4. Conclusions**

251 For the first time, a short-range structural analysis was performed on sodium
252 Taeniolite, which has allowed to shed a light on the distribution of the heteroatoms in
253 the framework, to estimate the actual layer charge, which is lower than the nominal one
254 and to propose a more accurate composition,
255 $\text{Li}_{0.91}\text{Na}_{0.52}[\text{Si}_8][\text{Mg}_{4.57}\text{Li}_{1.43}]\text{O}_{20}\text{F}_4 \cdot 2.18\text{H}_2\text{O}$.

256 Additionally, the thermogravimetric results indicate a low water content, which
257 is compatible with the presence of poor hydrated exchangeable interlayer sodium and
258 sodium partially coordinated by basal oxygen of the tetrahedral sheet of the silicate.

259 Those results allow a deeper knowledge of the real Na-Taeniolite structure and
260 hence, a better understanding of its promising properties.

261

262 **ASSOCIATED CONTENT**

263 **Supporting Information**

264 The Supporting Information is available free of charge on the ACS Publications webs.

265 Figures include the K-Taeniolite and Protoamphibole structures and the DTG curve of
266 Na-TAE.

267

268 **AUTHOR INFORMATION**

269 **Corresponding Author**

270 *E-mail: alba@icmse.csic.es

271 **Notes**

272 The authors declare no competing financial interest.

273

274 **ACKNOWLEDGEMENT**

275 The authors would like to thank the Junta de Andalucía (Spain) and FEDER (Proyecto
276 de Excelencia de la Junta de Andalucía, project P12-FQM-567), to the Spanish State
277 Program R+D +I oriented societal challenges and FEDER (Project MAT2015-63929-R)
278 and ENRESA (contract nº 0079000237) for financial support. F.J. Osuna thanks his
279 grant to the training researcher program associated to the excellence project of Junta de
280 Andalucía (P12-FQM-567). Finally, Dr. Pavón thanks her grant to Andalucía Talent
281 Hub Program, co-funded by the EU in 7FP, Marie Skłodowska-Curie actions (nº
282 291780) and the Junta de Andalucía.

283

284
285
286
287
288
289

Table 1.

Fit parameters of NMR spectra of Na-TAE.

	δ (ppm)	Fwhh (kHz)	%	assignment
²⁹ Si	-83.39	0.20	3.5	Q ² (protoamphibole)
	-85.07	0.12	1.3	Q ² (protoamphibole)
¹⁹ F	-91.37	0.18	95.2	Q ³ (OAl)
	-177.39	1.07	32.6	F(Mg-Mg-Mg)
	-183.53	1.42	63.4	F(Mg-Mg-Li)
²³ Na	-189.44	1.06	4.0	F(Mg-Li-Li)
	-8.86	0.70	6.8	Fully hydrated
	-12.97	1.98	71.6	
	-31.35	1.95	15.2	Poor hydrated
⁶ Li	-49.53	1.61	6.4	
	0.23	0.06	39.0	Interlayer
	-0.72	0.03	61.0	Framework

290 **FIGURE CAPTION**

291 **Fig. 1.** XRD of Na-TAE. *=protoamphibole=00-013-0409

292 **Fig. 2.** SEM image and EDX spectra of Na-TAE: a) particles with layered morphology,
293 and, b) particles with compacted morphology.

294 **Fig. 3.** IRFT spectrum of Na-TAE. The vibrational peaks have been labelled with the
295 vibrations mode.

296 **Fig. 4.** ^{29}Si , ^{19}F , ^6Li and ^{23}Na MAS NMR spectra of Na-TAE.

297 **Fig. 5.** Chemical shift and absolute value of layer charge correlation of trioctahedral 2:1
298 phyllosilicates as a function of structural parameters, referred to O_{20}F_4 units. Symbols
299 represents the experimental data from literature (Alba et al., 2006; Gerstmans et al.,
300 2008; Keenan et al., 2013; Sanz et al., 2006), *oh* is the composition of the octahedral
301 sheet, *i.s.* is the composition of the interlayer space and lines are the data fittings: blue
302 line (line I), cyan line (line II) and purple line (line III).

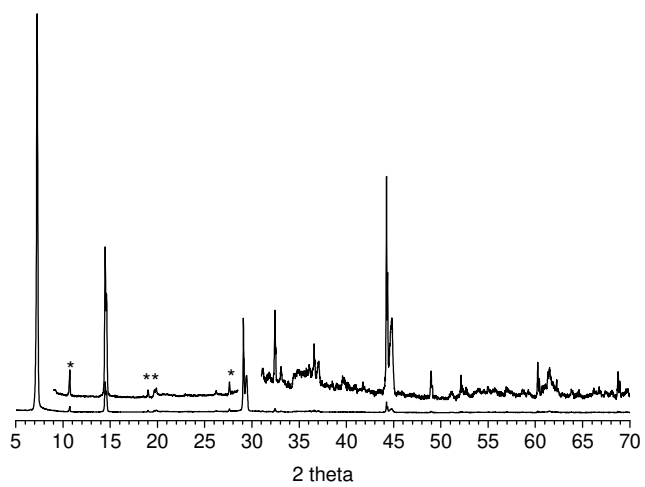
303

304

305

306 **Fig. 1**

307



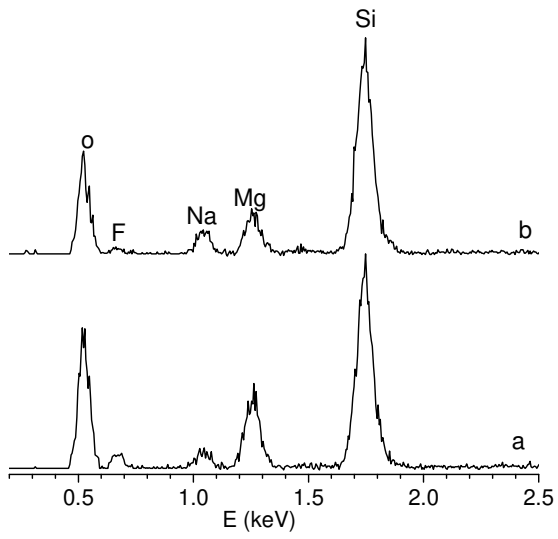
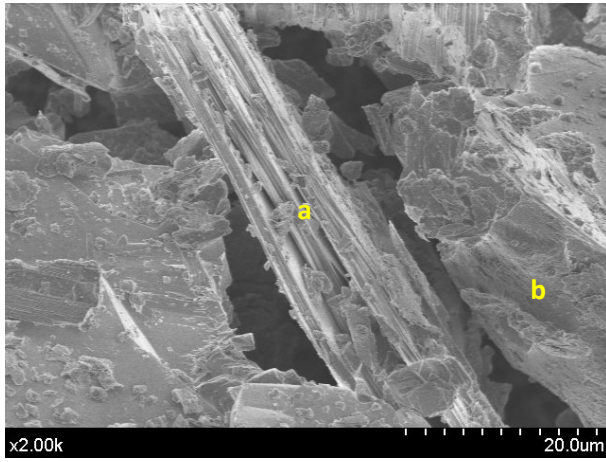
308

309

310 **Fig. 2**

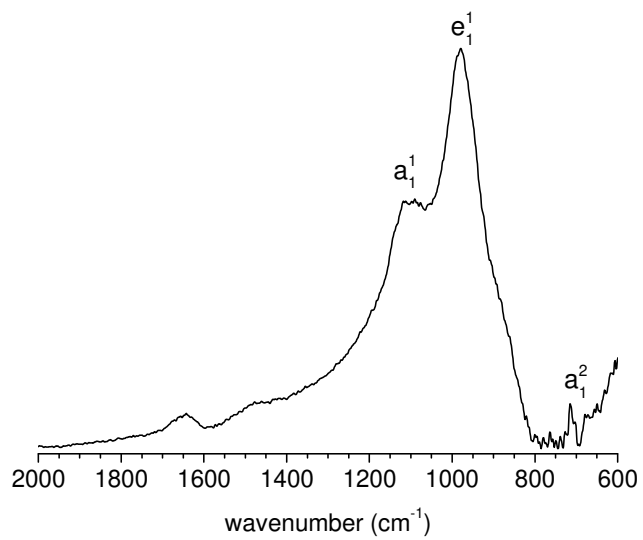
311

312



313 **Fig. 3**

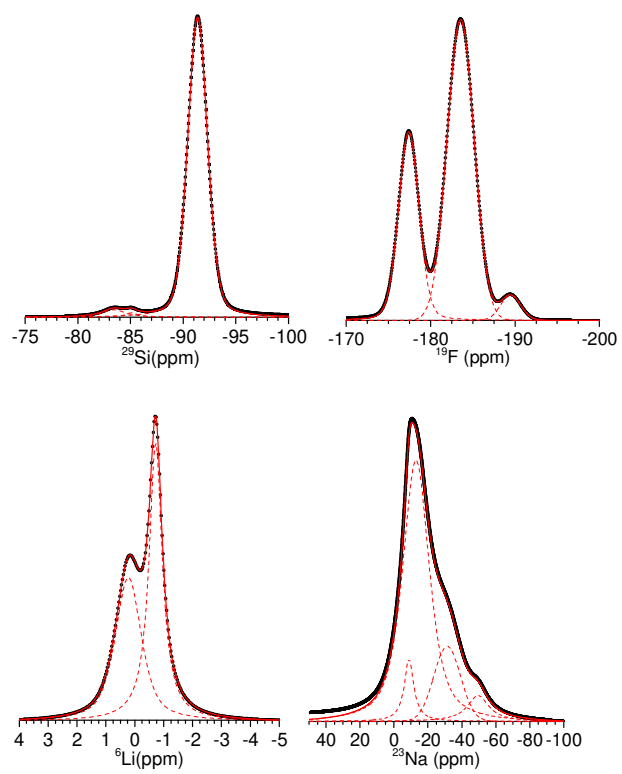
314



315

316 **Fig. 4**

317



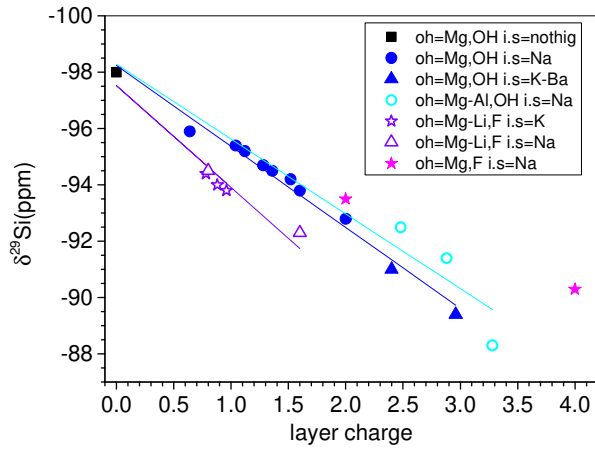
318

319

320 **Fig. 5**

321

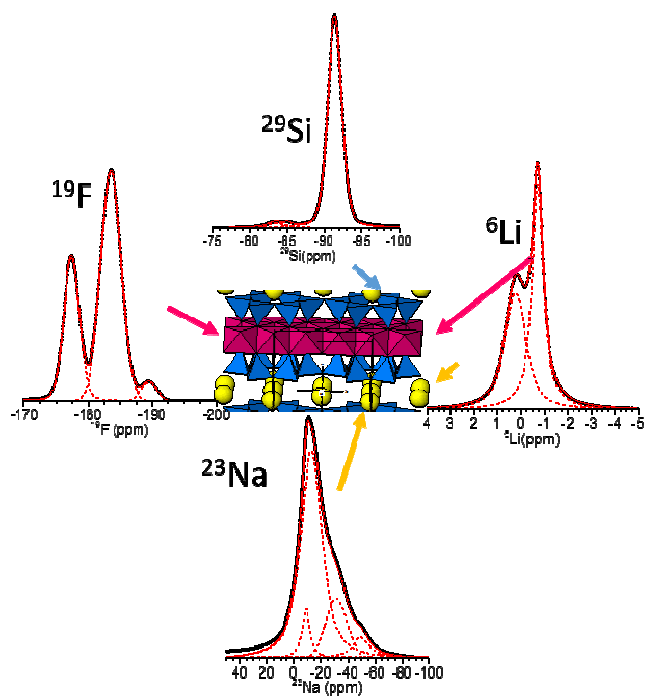
322



323

324

325 **GRAPHICAL ABSTRACT**



326

327 **REFERENCE**

328 Alba, M.D., Becerro, A.I., Castro, M.A., Perdigon, A.C., 2001a. Hydrothermal
329 reactivity of Lu-saturated smectites: Part I. A long-range order study. *Am. Miner.* 86,
330 115-123.

331 Alba, M.D., Becerro, A.I., Castro, M.A., Perdigon, A.C., 2001b. Hydrothermal
332 reactivity of Lu-saturated smectites: Part II. A short-range order study. *Am. Miner.* 86,
333 124-131.

334 Alba, M.D., Castro, M.A., Chain, P., Orta, M.M., Pazos, M.C., Pavon, E., 2010.
335 HYDROTHERMAL STABILITY OF LAYERED SILICATES IN NEUTRAL AND
336 ACIDIC MEDIA: EFFECT ON ENGINEERED-BARRIER SAFETY. *Clay Clay Min.*
337 58, 501-514.

338 Alba, M.D., Castro, M.A., Naranjo, M., Pavon, E., 2006. Hydrothermal reactivity
339 of Na-n-micas (n=2, 3, 4). *Chem. Mat.* 18, 2867-2872.

340 Alba, M.D., Chain, P., Orta, M.M., 2009. Chemical reactivity of argillaceous
341 material in engineered barrier Rare earth disilicate formation under subcritical
342 conditions. *Appl. Clay Sci.* 43, 369-375.

343 Alves, H.J., da Rocha, A.M., Monteiro, M.R., Moretti, C., Cabrelon, M.D.,
344 Schwengber, C.A., Milinsk, M.C., 2014. Treatment of clay with KF: New solid catalyst
345 for biodiesel production. *Appl. Clay Sci.* 91-92, 98-104.

346 Barrer, R.M., 1989. SHAPE-SELECTIVE SORBENTS BASED ON CLAY-
347 MINERALS - A REVIEW. *Clay Clay Min.* 37, 385-395.

348 Baumgartner, A., Sattler, K., Thun, J., Breu, J., 2008. A route to microporous
349 materials through oxidative pillaring of micas. *Angewandte Chemie - International*
350 *Edition* 47, 1640-1644.

351 Bond, S.P., Gelder, A., Homer, J., McWhinnie, W.R., Perry, M.C., 1991. LI-6
352 MAGIC ANGLE SPINNING NUCLEAR-MAGNETIC-RESONANCE
353 SPECTROSCOPY - A POWERFUL PROBE FOR THE STUDY OF LITHIUM-
354 CONTAINING MATERIALS. *J. Mater. Chem.* 1, 327-330.

355 Breu, J., Raj, N., Catlow, C.R.A., 1999. Chiral recognition among trisdiimine-metal
356 complexes, Part 4. Atomistic computer modeling of Ru(bpy)(3) (2+) and Ru(phen)(3)
357 (2+) intercalated into low charged smectites. *J. Chem. Soc.-Dalton Trans.*, 835-845.

358 Breu, J., Seidl, W., Stoll, A.J., Lange, K.G., Probst, T.U., 2001. Charge
359 homogeneity in synthetic fluorohectorite. *Chem. Mat.* 13, 4213-4220.

360 Centi, G., Perathoner, S., 2008. Catalysis by layered materials: A review.
361 *Microporous Mesoporous Mat.* 107, 3-15.

362 Chain, P., Cota, A., El Mrabet, S., Pavon, E., Pazos, M.C., Alba, M.D., 2013.
363 Evaluation of rare earth on layered silicates under subcritical conditions: Effect of the
364 framework and interlayer space composition. *Chem. Geol.* 347, 208-216.

365 Cheng, S., 1999. From layer compounds to catalytic materials. *Catal. Today* 49,
366 303-312.

367 Garcia-Jimenez, M.J., Cota, A., Osuna, F.J., Pavon, E., Alba, M.D., 2016. Influence
368 of temperature and time on the Eu³⁺ reaction with synthetic Na-Mica-n (n=2 and 4).
369 *Chem. Eng. J.* 284, 1174-1183.

370 Gerstmans, A., Urbanczyk, L., Jerome, R., Robert, J.L., Grandjean, J., 2008. XRD
371 and NMR characterization of synthetic hectorites and the corresponding surfactant-
372 exchanged clays. *Clay Min.* 43, 205-212.

373 Gibbs, G.V., 1969. Crystal structure of protoamphibole. *Mineral. Soc. Amer. Spec.*
374 *Pap.* 2, 101-109.

375 Gibbs, G.V., Bloss, F.D., Shell, H.R., 1960. PROTO-AMPHIBOLE, A NEW
376 POLYTYPE. *Am. Miner.* 45, 974-989.

377 Gregorkiewitz, M., Rausell-Colom, J.A., 1987. CHARACTERIZATION AND
378 PROPERTIES OF A NEW SYNTHETIC SILICATE WITH HIGHLY CHARGED
379 MICA-TYPE LAYERS. *Am. Miner.* 72, 515-527.

380 Huve, L., Delmotte, L., Martin, P., Ledred, R., Baron, J., Saehr, D., 1992. F-19
381 MAS-NMR STUDY OF STRUCTURAL FLUORINE IN SOME NATURAL AND
382 SYNTHETIC 2/1 LAYER SILICATES. *Clay Clay Min.* 40, 186-191.

383 Kalo, H., Milius, W., Brau, M., Breu, J., 2013. Synthesis and single crystal
384 structure refinement of the one-layer hydrate of sodium brittle mica. *J. Solid State*
385 *Chem.* 198, 57-64.

386 Kalo, H., Moller, M.W., Ziadeh, M., Dolejs, D., Breu, J., 2010. Large scale melt
387 synthesis in an open crucible of Na-fluorohectorite with superb charge homogeneity and
388 particle size. *Appl. Clay Sci.* 48, 39-45.

389 Kaviratna, H., Pinnavaia, T.J., 1994. ACID-HYDROLYSIS OF OCTAHEDRAL
390 MG²⁺ SITES IN 2/1 LAYERED SILICATES - AN ASSESSMENT OF EDGE
391 ATTACK AND GALLERY ACCESS MECHANISMS. *Clay Clay Min.* 42, 717-723.

392 Keenan, C.D., Herling, M.M., Siegel, R., Petzold, N., Bowers, C.R., Rössler, E.A.,
393 Breu, J., Senker, J., 2013. Porosity of Pillared Clays Studied by Hyperpolarized Xe-129
394 NMR Spectroscopy and Xe Adsorption Isotherms. *Langmuir* 29, 643-652.

395 Kitajima, K., Daimon, N., 1972. STUDIES ON SWELLING OF NA-
396 TAENIOLITE WITH WATER. *Chem. Lett.*, 953-956.

397 Kitajima, K., Koyama, F., Takusagawa, N., 1985. SYNTHESIS AND SWELLING
398 PROPERTIES OF FLUORINE MICAS WITH VARIABLE LAYER CHARGES. *Bull.*
399 *Chem. Soc. Jpn.* 58, 1325-1326.

400 Kitajima, K., Sugimori, K., Daimon, N., 1973. STUDIES OF SWELLING OF NA-
401 TAENIOLITE WITH WATER. Nippon Kagaku Kaishi, 1885-1892.

402 Kitajima, K., Takusagawa, N., 1990. EFFECTS OF TETRAHEDRAL
403 ISOMORPHIC SUBSTITUTION ON THE IR-SPECTRA OF SYNTHETIC
404 FLUORINE MICAS. Clay Min. 25, 235-241.

405 Kitajima, K., Taruta, S., Takusagawa, N., 1991. EFFECTS OF LAYER CHARGE
406 ON THE IR-SPECTRA OF SYNTHETIC FLUORINE MICAS. Clay Min. 26, 435-
407 440.

408 Komarneni, S., Ravella, R., Park, M., 2005. Swelling mica-type clays: synthesis by
409 NaCl melt method, NMR characterization and cation exchange selectivity. J. Mater.
410 Chem. 15, 4241-4245.

411 Kondo, R., Daimon, M., Asaga, K., Nishikawa, T., Kitajima, K., Daimon, N., 1980.
412 SOLID-SOLUTIONS IN THE SYSTEM TAENIOLITE-SODIUM TAENIOLITE
413 AND THEIR PROPERTIES. J. Am. Ceram. Soc. 63, 41-43.

414 Kunz, D.A., Schmid, J., Feicht, P., Erath, J., Fery, A., Breu, J., 2013. Clay-Based
415 Nanocomposite Coating for Flexible Optoelectronics Applying Commercial Polymers.
416 ACS Nano 7, 4275-4280.

417 Labouriau, A., Kim, Y.W., Chipera, S., Bish, D.L., Earl, W.L., 1995. A F-19
418 nuclear magnetic resonance study of natural clays. Clay Clay Min. 43, 697-704.

419 Lagaly, C., 1994. Layer Charge Determination by Alkylammonium Ions., in:
420 Society, C.M. (Ed.), Layer Charge Characteristics of 2:1 Silicates Clay Minerals. CMS
421 Workshop Lecture, Boulder Colorado, pp. 2-46.

422 Lezhnina, M., Benavente, E., Bentlage, M., Echevarría, Y., Klumpp, E., Kynast, U.,
423 2007. Luminescent Hybrid Material Based on a Clay Mineral. Chem. Mat. 19, 1098-
424 1102.

425 Mallouk, T.E., Gavin, J.A., 1998. Molecular recognition in lamellar solids and thin
426 films. *Accounts Chem. Res.* 31, 209-217.

427 Mariychuk, R., Baumgartner, A., Wagner, F.E., Lerf, A., Dubbe, A., Moos, R.,
428 Breu, J., 2007. Synthesis, structure, and electric conductivity of ferrous tainiolite and its
429 oxidative conversion into coarse-grained swellable smectite. *Chem. Mat.* 19, 5377-
430 5387.

431 Massiot, D., Fayon, F., Capron, M., King, I., Le Calve, S., Alonso, B., Durand,
432 J.O., Bujoli, B., Gan, Z.H., Hoatson, G., 2002. Modelling one- and two-dimensional
433 solid-state NMR spectra. *Magn. Reson. Chem.* 40, 70-76.

434 Mercier, L., Pinnavaia, T.J., 1998. A functionalized porous clay heterostructure for
435 heavy metal ion (Hg^{2+}) trapping. *Microporous Mesoporous Mat.* 20, 101-106.

436 Miyake, M., Suzuki, T., Suzuki, T., 1993. CATION-EXCHANGE SELECTIVITY
437 FOR K^+ , Mg^{2+} , AND Ca^{2+} IONS ON SODIUM-SUBSTITUTED TAENIOLITE.
438 *Chem. Mat.* 5, 1327-1331.

439 Moller, M.W., Hirsemann, D., Haarmann, F., Senker, J., Breu, J., 2010. Facile
440 Scalable Synthesis of Rectorites. *Chem. Mat.* 22, 186-196.

441 Moore, G.J., Zavalij, P.Y., Whittingham, M.S., 1997. Fluorophlogopite and
442 taeniolite: Synthesis and nanocomposite formation, in: Komarneni, S., Parker, J.C.,
443 Wollenberger, H.J. (Eds.), *Nanophase and Nanocomposite Materials II*. Materials
444 Research Society, Warrendale, pp. 501-506.

445 Muller, F., Besson, G., Manceau, A., Drits, V.A., 1997. Distribution of
446 isomorphous cations within octahedral sheets in montmorillonite from Camp-Bertaux.
447 *Phys. Chem. Miner.* 24, 159-166.

448 Naranjo, M., Castro, M.A., Cota, A., Osuna, F.J., Pavon, E., Alba, M.D., 2015.
449 Synthesis temperature effect on Na-Mica-4 crystallinity and heteroatom distribution.
450 Microporous Mesoporous Mat. 204, 282-288.

451 Okada, T., Ogawa, M., 2017. Adsorbents Derived from Layered Solids, in: Nakato,
452 T., Kawamata, J., Takagi, S. (Eds.), Inorganic Nanosheets and Nanosheet-Based
453 Materials: Fundamentals and Applications of Two-Dimensional Systems. Springer
454 Japan, Tokyo, pp. 263-301.

455 Park, M., Lee, D.H., Choi, C.L., Kim, S.S., Kim, K.S., Choi, J., 2002. Pure Na-4-
456 mica: Synthesis and characterization. Chem. Mat. 14, 2582-2589.

457 Paulus, W.J., Komarneni, S., Roy, R., 1992. BULK SYNTHESIS AND
458 SELECTIVE EXCHANGE OF STRONTIUM IONS IN NA4MG6AL4SI4O20F4
459 MICA. Nature 357, 571-573.

460 Pavon, E., Castro, M.A., Cota, A., Osuna, F.J., Pazos, M.C., Alba, M.D., 2014.
461 Interaction of Hydrated Cations with Mica-n (n=2, 3 and 4) Surface. J. Phys. Chem. C
462 118, 2115-2121.

463 Pavon, E., Castro, M.A., Naranjo, M., Orta, M.M., Pazos, M.C., Alba, M.D., 2013.
464 Hydration properties of synthetic high-charge micas saturated with different cations: An
465 experimental approach. Am. Miner. 98, 394-400.

466 Rayner-Canham, A.G., Overton, T., 2006. Descriptive Inorganic Chemistry, 4th ed.
467 Freeman, New York.

468 Sanz, J., Robert, J.L., Diaz, M., Sobrados, I., 2006. Influence of charge location on
469 Si-29 NMR chemical shift of 2 : 1 phyllosilicates. Am. Miner. 91, 544-550.

470 Schollhorn, R., 1996. Intercalation systems as nanostructured functional materials.
471 Chem. Mat. 8, 1747-1757.

472 Schoonheydt, R.A., Umemura, Y., 2017. Clay Minerals as Natural Nanosheets, in:
473 Nakato, T., Kawamata, J., Takagi, S. (Eds.), Inorganic Nanosheets and Nanosheet-
474 Based Materials: Fundamentals and Applications of Two-Dimensional Systems.
475 Springer Japan, Tokyo, pp. 33-53.

476 Sposito, G., Prost, R., 1982. STRUCTURE OF WATER ADSORBED ON
477 SMECTITES. Chem. Rev. 82, 553-573.

478 Suzuki, N., Komuro, T., Kanzaki, Y., 2008. Interfering effect for the cesium ion-
479 exchange property of sodium difluorotetrasilicate and sodium taeniolite by alkaline-
480 earth metal ions. Bull. Chem. Soc. Jpn. 81, 912-916.

481 Thomas, J.M., 1988. UNIFORM HETEROGENEOUS CATALYSTS - THE ROLE
482 OF SOLID-STATE CHEMISTRY IN THEIR DEVELOPMENT AND DESIGN.
483 Angew. Chem.-Int. Edit. 27, 1673-1691.

484 Tomohiko, O., Yoko, S., Makoto, O., 2014. Designed nanostructures of clay for
485 controlled adsorption of organic compounds. Journal of Nanoscience and
486 Nanotechnology 14, 2121-2134.

487 Toraya, H., Iwai, S., Marumo, F., Hirao, M., 1977. CRYSTAL-STRUCTURE OF
488 TAENIOLITE, KLIMG₂SI₄O₁₀F₂. Z. Kristall. 146, 73-83.

489 Vaccari, A., 1999. Clays and catalysis: a promising future. Appl. Clay Sci. 14, 161-
490 198.

491 Welch, M.D., Liu, S.X., Klinowski, J., 1998. Si-29 MAS NMR systematics of
492 calcic and sodic-calcic amphiboles. Am. Miner. 83, 85-96.

493 Yamaguchi, T., Ikuta, K., Taruta, S., Kitajima, K., 2012. Morphology control and
494 interlayer pillaring of swellable Na-taeniolite mica crystals. Mater. Sci. Eng. B-Adv.
495 Funct. Solid-State Mater. 177, 524-527.

496