**Tailoring structured WGS catalysts: Impact of multilayered concept on the water surface interactions**

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**Abstract**

A novel multilayered approach for designing structured WGS catalyst is employed in this study as a response to the lack of new strategies in the literature. The approach proposes the use of two successive layers with different functionalities on metallic micromonolith substrate. The WGS catalyst behavior is modulated by the nature of the inner layer which determines the active species surface population by acting on the water activation step. So, the catalytic promotion attained by introducing inner ceria containing solids with increasing number of oxygen defects is intensely analyzed through FT-IR and H2O-TPD. Several evidences about the participation of the oxygen vacancies, as key sites, for water absorption processes are established. Besides, remarkable relationships between the water absorption strengths and the water splitting processes within their influence on the catalyst performances are also discussed.

***Keywords*** Structured WGS catalysts; Ceria oxygen vacancies; Water splitting; Water Absorption; H2O-TPD

1. **Introduction**

Among the new green energies, H2 appears as promissory vector. One of the requirements to its success lies in the removal of CO coproduced with H2 in the reformate process in order to avoid the poisoning of Pt-based anode of the Polymeric Exchange Membranes Fuel Cells (PEMFC). The water gas shift (WGS, CO + H2O 🡪CO2 + H2) reaction allows CO removal at the same time that increases the H2 concentration. The reaction is also moderately exothermic and equilibrium limited and therefore, the control of the temperature is a crucial task [1]. Moreover, WGS reaction is characterized by high contact times and as a result, high reactor volume necessary to achieve good catalytic performances [2].

The industrially established WGS process is not well suited for small-scale applications, such as residential fuel cells or on-board hydrogen generators, for which reduced volume and weight catalytic beds are required. In fact, on-board portable applications intrinsically demand long lasting and highly active catalytic systems which should also support fast start-up and shutdown conditions. Attending to the mobile device power requirements, the drawbacks presented by the traditional WGS catalysts, e.g. slow response to transient operation of flow rate and temperature, could be overpassed applying noble metal (NM) based catalysts [1,3]. Within the reported NM catalysts, those based on Pt formulations appear as the best solutions especially when coupled with ceria containing support [4–7]. The use of ceria presumably intensifies the rate of WGS CO conversion by increasing the water activation rate due to its advantageous electronic properties which are, in turn, intimately associated to its structural defects, i.e. oxygen vacancies (Ov) [8,9]. So, the attention received by the oxygen defects lies on its capacity for dissociating water species and on the concomitant enhancement of the WGS catalyst’ reaction rates [10,11]. A possible way to control and increase the number of oxygen defect sites within ceria matrix is its doping with trivalent metal in proportion where the formation of solid solution is observed [12–18].

Nevertheless, not only the catalyst but also the reactor design is of crucial importance for the WGS process on-board applications [2]. An optimal structured reactor design should solve practical problems arising from WGS reaction particularities. The desired performance criteria for catalysts bed are low pressure drop, high catalyst loading per reactor volume, high external fluid to catalyst mass and heat transfer rates and high internal mass transfer rate within the catalyst [19–22]. In this context, parallel channel metallic micromonoliths appear as an attractive alternative from their inherent high thermal conductivities which assure enhanced heat transfer managements and allow shorter contact times, reduced transport limitations and higher catalyst surface-to-volume ratios. So, metallic micromonoliths should lead to compact catalytic beds and, then, to increased overall process efficiencies.

In a previous work [20], the oxygen vacancies and metallic micromonoliths benefits were merged on the development of a novel structured catalyst composed by two consecutives layers with different tasks: an inner layer of material able to absorb water and release activated water species in continuous manner and an outer catalyst layer where the WGS reaction should occur. The greater WGS overall reaction rate in presence of the inner layer was related to superior apparent H2O partial kinetic orders and consequently, to higher H/OH species population in the vicinity of the Pt active sites.

From the introduction of the multilayered concept, this work provides an intense analysis of the role of the buffer layer as a function of its water splitting capacity. The studied structured catalytic systems are schematically presented in Figure 1. For that purpose, Pt(2%)/CeO2/Al2O3 was maintained as catalyst for the all structured catalyst outer layers meanwhile the nature of the inner layer, named buffer layer, was varied. Three buffer materials differing on their water splitting capacity were chosen:

1. ɣ-Al2O3 chosen for its chemical similarity to oxidized stainless steel monolith surface and more importantly because of low available number of oxygen defect sites for water activation. In fact, this material is generally considered as inactive support for WGS reaction [32, 33];
2. CeO2/Al2O3 chosen as material with similar textural properties to that of ɣ-Al2O3 but with increased number of oxygen defect sites due to the presence of CeO2 surface layer;
3. Ce0.9Eu0.1O2/Al2O3 chosen for its expected intensification of the oxygen vacancies concentration due to Eu3+ to Ce4+ substitution and solid solution formation accompanied by oxygen defects formation via charge compensation mechanism [18]. In addition, Eu2O3 do not possess by itself any intrinsic activity for the WGS reaction.

Hence, as an initial hypothesis, one could expect that the WGS activity sequence should correlate directly to buffer’ ability to activate water, presumably in the following descending activity order Ce0.9Eu0.1O2/Al2O3 > CeO2/Al2O3 > ɣ-Al2O3. The use of Pt(2%)/CeO2/Al2O3 catalyst prepared in one unique batch with equivalent properties (chemical composition, metal dispersion etc.) consent ascribing the observed changes uniquely to the buffer layer’ role.

Therefore, this study offers a tuning procedure for enhanced WGS structured catalysts at the same time that delivers a comprehensive analysis of the water ceria surface interactions which, depending on defect surface population, tailor the WGS catalytic behaviors. In fact, the application of the multilayer approach pretends to prove the design suitability and to highlight it as a potential strategy for achieving advanced structured WGS catalysts.

# Materials and Methods

* 1. *Preparation of metallic micromonolith’ substrates:*

The micromonoliths were prepared by rolling up flat and corrugated ferritic stainless steel a foil (Fecralloy) around a spindle. The resulting structures are cylinders of 1.6 cm diameter, 3 cm height and 540 cm2 geometric surface area with cell density of 2067 cpsi. The micromonoliths were thermally pretreated during 22h at 900ºC in order to generate α-Al2O3 oxidized surface layer for better catalyst adhesion during the deposition assured by roughness increase for physical anchorage and chemical compatibility [23,24].

* 1. *Catalyst and buffer´s synthesis*:

The CeO2/Al2O3 and the Ce0.9Eu0.1O2/Al2O3 buffer materials were prepared by wet impregnation of their respective nitrates over a commercial ɣ-Al2O3 (Sasol). The CeO2 (Ce0.9Eu0.1O2):Al2O3 ratio was fixed to 20:80 (wt.%) and the impregnation was carried out in ethanol solution, posteriorly evaporated under reduced pressure. The obtained dry solid was treated with diluted NH3 solution during 30 min, washed, filtrated and calcined at 450oC for 4h at 10oC/min heating rate. The as prepared buffers are denominated Al, CeAl and CeEuAl for Al2O3,CeO2/Al2O3 and Ce0.9Eu0.1O2/Al2O3 respectively.

The platinum catalyst was prepared in a similar way by wetness impregnation of the targeted Pt loading of 2wt.% and using tetrammoniun nitrate platinate (Johnson Mattey) as metal precursor. As catalyst support a commercial Puralox CeO2/Al2O3 (Sasol) was used. After impregnation, the catalyst was dried and calcined at 350oC during 8h with a heating rate of 5oC·min-1. The obtained catalyst was labelled Pt/CeAl.

* 1. *Preparation of the structured catalysts:*

The washcoating method was used to cover the metallic substrate surface with buffer and catalyst layer successively. For this, slurries with adjusted rheological properties were prepared. For all employed solids, prepared suspensions contain 18.14 wt.% of solid, 1.96 wt.% of polyvinylalcohol (PVA) solution (1.5 wt.% in water), 17.65 wt.% of colloidal Al2O3 suspension (Nyacol, 20 wt.% in water) and 62.25 wt.% of H2O. Prior use all solids were grinded and sieved to particle size lower than 10 µm. During the coating process the PVA, was used to facilitate the suspension impregnation by capillarity within the channels meanwhile Al2O3 Nyacol helps the stabilization of the catalyst particles in the resulting suspension [23].

The micromonoliths were immersed and then removed from suspensions always at a constant rate of 2 cm·min-1 for 1 min. Centrifugation combined with air flushing was employed to remove the excess suspension and to avoid the channels obstruction. For the buffer layer the procedure was repeated several times till the desired amount of 300 mg approx. was deposited. The micromonoliths were calcined at 450oC during 4h at 2oC·min-1 heating ramp. The remaining suspensions after layer deposition were dried and calcined in the same conditions as micromonoliths and use for comparison between the powder catalysts. The chosen nomenclature for the suspensions were S\_Al, S\_CeAl and S\_CeEuAl, for the suspensions resulting from the buffers based on Al, CeAl and CeEuAl solids, respectively.

Once a homogeneous buffer layer obtained, the catalyst layer was coated in the same proportions and conditions of deposition used for the buffers. Finally, the micromonoliths were calcined at 350oC for 8h at 2oC·min-1 heating ramp. The structured catalysts constituted by two layers, buffer and catalyst ones are labelled M\_Al, M\_CeAl and M\_CeEuAl, while the buffer only deposited monoliths are labelled Mb\_Al, Mb\_CeAl and Mb\_CeEuAl (with “b” reffered to buffer only), respectively. The targeted quantity of catalyst of approx. 300 mg was achieved after successive washcoating steps. The actual achieved amounts are shown in Table 1.

For all layered systems, both buffer and external catalyst´ layer are maintained constant with the only variation being the chemical composition of the buffer layer. Both layers present similar density and textural properties (all having at least 80% of alumina) and, from the similar deposited quantities, result in similar layer thickness. The layer thickness was estimated using micromonolith total geometrical area, catalyst density and total pore volume (Table 1).

* 1. *Characterization technique:*

The elemental composition of the powder samples (buffers, catalysts and suspensions) was analyzed by X-Ray fluorescence (XRF) spectrometry carried out in an X Panalytical AXIOS PW4400 using Rh tube as a source of radiation. The textural properties were studied by N2 adsorption-desorption measurements at liquid nitrogen temperature in a Micrometrics ASAP 2010 instrument. Before analysis, the samples were degassed for 2h at 150ºC in vacuum. For the monoliths, especially designed homemade sample holder was used allowing the analysis of whole monolith.

X`Pert Pro PANalytical instrument was used for X-ray diffraction (XRD) analysis. Diffraction patterns were recorded using Cu Kα radiation (40 mA, 45kV) over a 2ϴ-range of 10 to 95º and a position-sensitive detector using a step size of 0.05º and a step time of 240 s.

Scanning electron microscopy (SEM) analysis and adherence test were performed. SEM was carried out on a JEOL 5400 microscope. The adherence of the layer to the substrate was evaluated introducing the monolith in acetone during 30 min under ultrasound treatment and adherence was related to the amount of catalyst lost during the test in respect to the initial loading.

FTIR spectra were recorded using THERMO NICOLET Avatar 380 FT-IR Spectrophotometer, equipped with a DTGS detector, spectral resolution of 4 cm−1 and 128 accumulated scans. The experiments were performed *in situ* using purpose-made IR cell connected to conventional vacuum adsorption apparatus with residual pressure lower than 10−5 mbar. The experiments consisted of water absorption in small doses at room temperature till saturation. After then, the sample was heated to 50oC, cooled down to room temperature and a spectrum was recorded. This procedure repeated also at 100, 150 and 200 oC. The changes produced in respect to the initial IR spectrum, before any water addition, were analyzed.

For the H2O-TPD, water pulses, 500 L each, were introduced to the sample at 120ºC using N2 as vapor carrier until surface saturation. Once water signal stabilized (m/z = 18), the sample was heated up to 600 ºC with 5 ºC.min-1 heating rate in order to desorb all water species. The outlet gas composition was normalized by the surface area of each sample and analyzed with mass spectrometer PFEIFFER Vacuum PrismaPlus controlled by Quadera software.

* 1. *Catalytic activity:*

The water gas shift reaction was carried out in tubular flow reactor at atmospheric pressure. All samples were tested in the 180–350°C temperature range, with gas composition, containing 7% CO, 9% CO2, 50% H2 and 30% H2O (balanced with N2), used to simulate the outlet of a typical ethanol reformer reactor. Constant space velocity (WHSV) of 80 L·gcat-1·h-1 was used for each micromonoliths test. The powder samples (PtCeAl and S-PtCeAl) were tested using sieved solids in the 600–800 μm fraction range. All samples were activated at 350°C during 2h in H2 stream (10 vol.%) prior the WGS reaction measurement. The CO and CO2 contents were analyzed with an ABB gas analyzer and the activity is expressed in terms of CO conversion.

The effect on the catalytic activity of water partial pressure variation was also studied at 310°C. In these experiments, the total flow was kept constant using N2 to maintain the GHSV. The composition was 50 % H2, 7 % CO, 3 % CO2, while the water concentration was modified taking values of 15%, 30% and 40% (v/v). The CO2 concentration decreased in these experiments in order to widen the range of available water partial pressures while maintaining the same GHSV.

1. **Results and discussion**

The chemical compositions of catalyst, buffers and their respective suspensions are presented in Table 2. All systems present compositions close to the desired loading, only minimal loss of CeO2 component in respect to alumina is observed. Nevertheless, the obtained Ce/Eu molar ratio of 12 instead of 9 in the originally designed solid indicates lower Eu incorporation within ceria matrix. Compared to the parent initial solids, the changes observed for the suspension’ chemical compositions could be attributed to the additional alumina component introduced during slurry preparation. The incorporation of colloidal alumina (Nyacol) increases overall alumina content and hence, automatically decreases ceria, europia and platinum loadings, as reflected in Table 2.

The textural properties of buffer solids, catalyst and respective suspensions are also shown in Table 2. All solids are mesoporous with specific surface area dominated by the presence of alumina. Among initial buffer solids, the pure alumina buffer sample presents the highest specific surface area diminishing respectively with CeO2 and/or Ce-Eu mixed oxide addition. In comparison to the corresponding bare solids, the textural properties of the suspensions are furthermore modified by the presence of Nyacol (192 m2/g; 0.58 cm3/g) as reflected in their higher specific surface areas and pore volumes.

All structured catalyst presented the same catalyst layer amount (Table 1). However, some slight differences are found for the buffer layer loadings between the samples. As for the powder samples, the structured ones containing alumna buffer layers (Mb\_Al and M\_Al) present the highest specific surface area which decrease with the addition of bare or mixed ceria oxide. No direct correlations on the minor variations of the specific surface area of the micromonoliths to the suspensions ones are found. Meanwhile the Mb\_Al and M\_Al specific surface areas increased in comparison to S\_Al, those of Mb(M)\_CeAl and Mb(M)\_CeEuAl decreased. Lower resistances against thermal sintering of the ceria based solids could account for the observed differences.

The XRD patterns of all powder samples are shown in Figure 2. Al, CeAl, CeEuAl and Pt/CeAl samples present all diffraction lines corresponding to CeO2 fluorite (JCPDS# 00-004-0593) and -Al2O3 (JCDPS# 00-048-0267) structures. In the case of CeEuAl, no diffraction lines due to crystalline europium oxide are detected. Moreover, the observed slight shifts in the CeO2 main diffractions in respect to those of CeAl indicate the formation of Ce-Eu mixed oxide. The calculation of ceria lattice parameter, on (111) main plane diffraction, shows an increase of the parameter value from 5.39Å for CeAl to 5.40 Å for CeEuAl sample accounting for partial Ce4+ substitution by Eu3+ ions. This substitution confirms the solid solution formation and implies also appearance of oxygen vacancies obtained by charge compensation mechanism (Ce4+ is substituted by greater ion with lower charge Eu3+)[1,2]. The lattice distortion also results in smaller CeO2 crystal size, as estimated by Scherrer`s equation, 38 nm for CeEuAl *vs.* 55 nm for CeAl. This fact could be attributed completely to the lattice alteration, produced by europium insertion, after which the crystal growth is restricted, resulting in smaller crystal domains. No diffractions attributed to Pt are observed for Pt/CeAl sample suggesting high dispersion of the noble metal with crystalline domains under the detection limit of the technique (5nm).

No important differences of suspension’s XRD patterns (Figure 2B) with respect to their parent solids are detected. The latter indicates the absence of solid structural changes during suspension preparation procedure. Still, the appearance of new alumina boehmite phase (JCDPS# 00-21-1307) corresponding to that issued from Nyacol is identified for the Pt/CeAl probably related to the lower calcination temperature in comparison to the buffer solids.

The adherence test within micromonolithic series indicates an excellent adherence of 94-97% catalyst layer remaining after the test. In all cases, the SEM micrographs confirm the presence of well-dispersed homogeneous catalytic layer as shown in Figure 3A representing selected micrograph obtained for M\_CeAl monolith and its corresponding cross section micrograph with included line analysis (Figure 3B). From left to right, three layers can be distinguished: i) the metallic substrate characterized by Fe-Cr presence; ii) a layer constituted by alumina produced during the thermal pretreatment of the monolith´s steel, the colloidal Al2O3 added during to the slurry preparation and the alumina component coming from the buffer itself and iii) an external surface layer rich in CeO2.

The powder Al, CeAl and CeEuAl samples, their corresponding suspensions (S\_Al, S\_CeAl and S\_ CeEuAl) and buffer only micromonoliths (Mb\_Al, Mb\_CeAl and Mb\_ CeEuAl) do not present catalytic activity at the employed temperature and gas mixture conditions. The bare catalysts powder and suspensions (Pt/CeAl and S\_Pt/CeAl) do not differ in activity indicating that the catalyst properties are not influenced by the catalyst to suspension switch operation (data not shown). In addition, the estimated layer thickness of a maximum of 7m (Table 1), indicates an effectiveness factor close to 1 for which the diffusional phenomena controlling the reaction can be ruled out [2,25,26]. Thus, all differences in activity for the multilayered samples will be related to the buffer layer nature[3]. The catalytic activity of all structured systems is shown in Figure 4. None of the catalysts achieves CO equilibrium conversion (Figure 4A), probably due to the exigent operational conditions employed in this study, i.e. low contact time and high H2 and CO partial pressures. Similarly, Kolb et al. [27][4] reported CO conversions below 30% at 310ºC employing similar feed streams and catalyst composition. Also, Germani et al. [28][5] presented comparable catalytic behavior for comparable feed stream composition and catalysts. The activity tendency in CO conversion terms obeys the following order: M\_Al < M\_CeAl < M\_CeEuAl. The activity trend is even more accentuated under normalization of the converted CO molecules over the overall specific surface area exposed to the reactants (Figure 4B). It is clear that the addition of ceria to alumina greatly improves the catalytic activity, improved even more with europium to ceria doping. One can postulate that the observed activity improvement is directly related to the concentration of oxygen vacancies during the reaction in a way that higher the vacancies number, higher the WGS activity. In addition, the covalency of the Ce-O bond is affected by the presence of Eu resulting in different capacity for facing up water dissociation process, i.e. different H/OH surface coverages from water dissociative reactions within solids oxygen vacancies.

The behavior of the micromonoliths as a function of buffer nature under different reaction atmospheres was also studied. In order to broaden the compositional range, the initial feed was slightly altered by reducing the CO2 concentration (7% CO + 3% CO2 + 50% H2 + 30% H2O). This allows to performing the composition variation study (one per experiment) under similar conditions. The observed activity variation is reflected in Figure 5.

No substantial changes upon CO2 concentration variation are observed. In fact, no important activity diminutions are observed until high CO2 concentrations in the feed stream are introduced being this resistance emphasized for both the ceria containing buffers. On the contrary, a notorious CO conversion decrement with the increase of hydrogen concentration was observed. Considering that WGS reaction is an equilibrium reaction, the incorporation of both products (CO2 or H2) should lead to decrease of the CO conversions in both cases. So, meanwhile a strong inhibition is detected from H2 species, CO2 molecules could be just considered as negative at higher gas concentrations. In very good agreement to the observed trends, for similar conditions and catalytic systems, kinetic equations with CO2 and H2 partial kinetic orders of 0 and -0.5 respectively, are proposed in literature [28,29].

The increment of CO concentration on the feed stream also entails a decrement of the CO conversions, related to the increase of the number of CO molecules, which have to be converted by the same number of active sites. Besides, the catalytic inhibition should be also related to the competitive CO-H2O absorptions over the active sites: Pt and Pt/Ceria interfaces. This means that, at higher CO coverages, the available OH population on the around the active sites should decrease and, as direct consequence, lead to lower CO conversions [30]. For those three components, the changes provoked by the feed variations remain within the same magnitude no matter the buffer nature and no important differences are observed. Besides and as expected, the ceria containing samples always present higher CO conversions.

The increase of H2O concentration (Figure 5D), however, involves significant changes in catalyst’s WGS behavior. At low water concentrations, all micromonoliths present the same conversion values gradually increasing with water enrichment of the feed. Since the WGS is an equilibrium, an enhancement of the activity with the water partial pressure rise is expected but this improvement appears to depends on the buffer`s composition following the sequence: M\_CeEuAl > M\_CeAl > M\_Al. Therefore, the sequence of activity correlates with the increasing oxygen vacancies concentration which favors and makes easier the water activation. The presence of buffer layer, associated to the existence of extra number of defect sites, results in changed electronic properties and influences the reactants coverages on the catalyst’ surface [25]. From the improved capacity for dissociating water, an increase of the OH/H species population near the active Pt sites could be related to the enhanced catalytic behaviors. According to the initial hypothesis, water activation capacity and consequent catalytic behavior can be modulated by the buffers composition in a way that higher the number ceria defects, higher the water activation rate and higher the activity.

The relationship between H2O surface interactions and buffer nature was evaluated by Infrared spectroscopy. In general, the introduction of certain water quantities to the system could lead either to water adsorption and/or to water dissociation. The Figure 6 presents the hydroxyl band evolution for the S\_Al, S\_CeAl and S\_CeEuAl samples during successive water introduction steps. The important alumina content in all the samples explains the observed similar spectra. For all the samples, the corresponding evolution with the water introduction for the 1764-1414 cm-1 region is also shown in Figure 6D. So, weak bands placed at about 1640-1595 cm-1 accounts for water molecularly adsorbed within different chemical environments, (HOH) [31]. For S\_Al (Figure 6A), different bands corresponding to hydroxyl species adsorbed on alumina surface and placed at 3781, 3734, 3692, 3580 and 3532 cm-1 dominates the spectrum. So, the first three bands are generally attributed to OH groups bounded mono or bicoordinated to tetrahedrally Al3+ species [32–34][7–9] and the last two ones (3580 and 3532 cm-1) are typical for H-bonded to OH species[7] or non-dissociated water molecules [31,35]. Successive water adsorptions clearly lead to 3734 cm-1 band intensity diminutions along to intensity increments of the molecularly water absorbed related bands (3580, 3532 and 1640 cm-1). Besides, the water doses entail very tiny increments of the 3781 and 3692 cm-1 OH bands indicating the minor water dissociation capacities characteristic of Al2O3 surfaces. In turn, in absence of important dissociative processes, increasing amounts of molecularly absorbed water species should provoke a decrease of the hydroxyl band intensity since the O-H bond strength will become weaker because of the OH – H2O dipole formation and interaction. So, despite that a slight Al2O3 surface capacities for dissociating water molecules cannot be completely ruled out, in agreement to the clear observed features the water introduction over S\_Al sample principally entails molecular water absorptions.

For S\_CeAl (Figure 6B), bands placed at 3777, 3731, 3687, 3580 and 3514 cm-1 are observed. As expected, some differences are detected on the band positions in respect to the initial Al bands accounting mainly for ceria presence and surface-water modified interactions. For instance, the OH band at 3781 cm-1 showed by S\_Al sample is shifted to 3777 cm-1 in the S\_CeAl one owing to different chemical environments provoked from the cerium oxide incorporation. Also, the S\_Al band placed at 3734 cm-1 is also slightly shifted to lower wavelengths probably due to the low intensity band cerium oxide contribution reported at 3710 cm-1 being attributed to OH(I) species adsorption [31][11]. The band situated at 3687 cm-1 is related to OH species on Ce3+ or Ce4+ atoms [35]. Again, the bands at ca. 3580 and 3514 cm-1 are associated undissociated water molecules [32,35,36][7,10]. So, subsequent water pulses entail superior Ce-OH band intensities (3731 and 3687 cm-1) suggesting the ceria surface ability for dissociating water species. The noticeable increment observed for the band at ca. 3687 cm-1 intends its relation to Ce3+ defective sites with its corresponding ability for dissociate water species. The S\_CeAl system, in comparison to the S\_Al one, exhibits more pronounced intensity growths on the bands associated to absorbed water molecules (3580, 3514 and 1640 cm-1) indicating that defective ceria surfaces also display higher water absorption capacities.

S\_CeEuAl sample upon water adsorption presents bands situated at 3736, 3690, 3571 and 3528 cm-1, respectively. An additional shift of the bands position is observed and tentatively correlated to different chemical environments achieved upon europium insertion into ceria lattice. No bands of europium oxide – H2O interaction are detected. The successive water introduction involves higher OH population (3736 and 3690 cm-1) on the S\_CeEuAl surface. Also, strongly pronounced band intensity raises are observed for the bands placed at 3571, 3528 and 1640 cm-1 evidencing its remarked water adsorption capacity. In parallel, the noticeably wide band at the 3490 to 2800 cm-1 range could be also directly related to higher absorbed water quantities.

Therefore, the determining role of the surface composition on the water dissociation capacities is clearly stablished from the OH band evolution as a function of the water partial pressures. The direct relationship between the water splitting ability and the oxygen vacancy population is evidenced since: higher the oxygen vacancies concentration, higher the water splitting capacity[12,13]. This water dissociation promotion could be explained considering the electrons intrinsically associated to the oxygen vacancies, Ce3+- Ov|e·. So, superior oxygen defect surface populations should lead to higher electronic densities which entail shifted Fermi’s level through favored electron delocalization processes and enhanced water reduction reactions [37,38]. Then, an intimate relationship between OH surface populations water splitting capacity and catalytic performances could be established [39,40].

On the other hand, interesting outcomes could be also extracted from the different spectra displayed for the samples when equal water amounts (3·10-18 mol/nm2) are introduced, Figure 7. So, not just the dissociated water species but also the water molecules amount absorbed on the surface responds to the following sequence: S\_CeEuAl > S\_CeAl > S\_Al. Thus, S\_CeEuAl sample absorbs the highest water amount on its surface at the same time that presents greatest water dissociative capacity. Hence, the oxygen defects concentration strongly influences the surface chemisorptive properties and results on modulated OH/H2O surface interactions and consequently, modulated catalytic behaviors [9].

After water saturation, the samples were degassed first by applying different pressures (and, then by heating at T = 50, 100, 150 y 200 ºC. For the 1780 – 1400 cm-1 region, the spectra obtained during the water desorption for S\_CeAl and S\_CeEuAl are shown in the Figure 8A. Apart from the described correlation between the oxygen vacancies population and the amount of water molecules absorbed, the surface defect concentration also affects to the absorption water strength since the S\_CeEuAl presents higher absorbed water amounts at the different temperatures evaluated during the desorption process.

The oxygen vacancies effect on the water surface interactions were also evaluated by H2O-TPD experiments over the S\_Al, S\_CeAl and S\_CeEuAl powder samples after surface saturation with successive water vapor pulses. The observed changes in this experiment should account for the different type of water surface interactions as a function of support nature. The m/z=18 profile under H2O-TPD conditions is presented in Figure 8B with the WGS temperature range (0-350 ºC) amplified. The desorptive profiles of the samples are composed by two main desorption zones as function of the temperature: 120-350ºC and 350-600ºC. The highest temperature desorption zone can be tentatively attributed to processes related to alumina presence around 500ºC and to the OH groups lost at temperature around 600ºC[14]. The desorption processes observed at lower temperature could be ascribed, in temperature ascending order, to physically adsorbed water at temperatures below 200ºC and to desorption of water adsorbed on ceria surfaces for temperatures above 200ºC. For ceria based samples, similar experimental H2O-TPD profiles were reported by Thinon et al.[15] [41]. According to their simulations, based on the kinetic law proposed by Germani et al. [28][5], the two main desorption processes at 120 ºC and 260 ºC should be respectively attributed to molecular and dissociative water desorptions.

So, the S\_Al desorption profile shows water desorptions earlier than in both ceria samples corroborating the stronger water interactions attained through the ceria incorporation. Besides, the low water amount desorbed about 220ºC suggests the tiny capacity for water splitting processes of Al2O3 surfaces. Also, the higher absorptive and dissociative capacities detected for both ceria samples are again underlined by the larger overall desorptive processes exhibited by those samples being much greater those showed by the S\_CeEuAl. The preferred water adsorptions over reduced ceria surfaces, more precisely, over Ce3+ sites associated to oxygen defects in pure ceria lattices account for the observed features [42].[16] The higher contribution discerned for the ceria based samples at about 260°C unequivocally ratifies the connection between oxygen vacancies populations with the superior water surface interaction strengths and also, with the enhanced water dissociative abilities. Therefore, the proposed catalytic activity sequence: M\_CeEuAl > M\_CeAl > M\_Al, which suits to the oxygen vacancies concentration within the inner layer of the structured catalysts, could be also extended to the different system abilities for interacting to the water species in terms of absorption, dissociation and interacting strength.

1. **Conclusions**

The confluence of the benefits provided by the use of metallic micromonolith and the multilayer approach for the WGS activity has been merged and deeply evaluated. The structured catalyst design includes the development of an innovative multilayered catalyst composed by an external PtCeAl catalyst and an internal buffer layer with varied chemical composition. The latter is finely tuned considering water dissociation as the reaction limiting step and the role of the oxygen defects concentration on it. The catalytic activity results evidenced that catalyst’ behavior is modulated by the buffer layer composition or, more precisely, by the increase of the number of oxygen vacancies. A superior oxygen vacancies concentration provokes higher electron densities and entails boosted water reductive abilities. Remarkably, in view of the FT-IR and H2O-TPD results, the defect surface concentration not just determines the dissociative water reactions by itself but also defines the molecular water interactions with the surface. The ensuing better abilities to absorb, maintain and dissociate the surface water species results on changed surface coverages, rising up the apparent partial H2O pressures and consequences on WGS catalytic improvements.

As a result, the multilayered concept for obtaining superior WGS catalytic performances is validated and intimately related to the changes on the water surface interactions. The proposed novel design for WGS structured catalysts appears as a potent strategy able to tailor the behavior of a standard WGS catalysts.

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